# New Structure of High-Temperature Zirconium Molybdate 

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#### Abstract

Zr}\left(\mathrm{MoO}_{4}\right)_{2}, M_{r}=411 \cdot 12\), crystallizes in two phases; high-temperature form is trigonal, $P \overline{3} 1 c, a$ $=10.1391(6), \quad c=11.7084(8) \AA, \quad V=1042.4(2) \AA^{3}$, $Z=6, D_{m}=3.91$ (4), $D_{x}=3.927 \mathrm{~g} \mathrm{~cm}^{-3}, \lambda\left(\mathrm{Mo} K a_{1}\right)$ $=0.70926 \AA, \mu=53.68 \mathrm{~cm}^{-1}, \quad F(000)=1128, T=$ $293 \mathrm{~K}, R=0.031$ for 855 independent reflexions with $F_{\text {obs }}>\sigma$. The zirconium atoms occupy two special positions and have octahedral coordination with normal $\mathrm{Zr}-\mathrm{O}$ bond lengths. The molybdenum atoms are in a distorted tetrahedral arrangement. The outstanding feature of this revised structure is the unusual short Mo-O(4) distance of $1.690 \AA$ with a bond strength of 1.89.


Introduction. This study is part of a systematic investigation of the correlation between the structure and the vibrational spectrum of molybdates and tungstates that contain a tetravalent cation. The $\mathrm{Zr}\left(\mathrm{MoO}_{4}\right)_{2}$ compound crystallizes in two different phases. The low-temperature form transforms reversibly into a high-temperature form at $\sim 913 \mathrm{~K}$ (Tarte \& Auray, 1982). The crystal structure of the parent compound $\mathrm{Hf}\left(\mathrm{MoO}_{4}\right)_{2}$ (Freundlich \& Thoret, 1967), which crystallizes only in the high-temperature form, was reported by Rimsky, Thoret \& Freundlich (1968). The structure was determined with only 81 reflexions for 23 variables. The reported bond distances do not match the atomic coordinates, some distances $[\mathrm{Mo}-\mathrm{O}(1)=1 \cdot 41, \mathrm{O}(1)-\mathrm{O}(2)=2 \cdot 17, \mathrm{Hf}(1)-\mathrm{O}(2)=$ $2.75 \AA$ ] do not agree with the distances predicted from Shannon's (1976) ionic radii, and a twelvefold coordination for $\mathrm{Hf}^{\mathrm{IV}}$ is most unlikely. Therefore we decided to carry out a conclusive detailed crystallographic redetermination of the title compound, $\mathrm{HT} \mathrm{Zr}\left(\mathrm{MoO}_{4}\right)_{2}$.

Experimental. Single crystals of $\mathrm{Zr}\left(\mathrm{MoO}_{4}\right)_{2}$ prepared by heating some of the powder sample $\left(\mathrm{ZrO}_{2}: 2 \mathrm{MoO}_{3}\right)$

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with an excess of $\mathrm{MoO}_{3}$. After 3 h at 1123 K the mixture was quenched, to inhibit generation of the low-temperature form. The crystals appeared as colourless plates. Precession photographs indicated Laue symmetry $\overline{3} m$ and systematic extinctions ( $h h 2 \bar{h} l$ for $l=2 n$ ) consistent with $P \overline{3} 1 c$. Cell dimensions obtained and refined from powder diffraction data, using Si as reference. Pycnometric density measured in toluene. Intensity data obtained on a Philips PW 1100 four-circle automatic diffractometer. Data-collection conditions: Mo $K \alpha_{1}$ with graphite monochromator, $\omega-2 \theta$ scan at $0.01^{\circ} \mathrm{s}^{-1}$ with $(1.5+$ $0.3 \tan ()^{\circ}$ scan range. Three control reflexions (630, $0,0,12, \overline{6} \overline{3} 0$ ) measured every hour showed relative standard deviation of $0 \cdot 008.1020$ independent reflexions measured up to $2 \theta_{\text {max }}=60^{\circ}$ ( $h: 0$ to $12, k: 0$ to 7 , $l:-16$ to 16), corrected for background, Lorentz and polarization effects, which provided 762 reflexions satisfying $F_{\text {obs }}>3 \sigma$. Absorption corrections with de Meulenaer \& Tompa's (1965) analytical method, program AGNOST (Ahmed, 1974). Distances along $a$, $b$ and $c$ were respectively $0.316,0.448$ and 0.100 mm . Min. and max. absorption correction factors 0.673 and $0.803 . \mathrm{Zr}^{4+}, \mathrm{Mo}^{6+}$ (International Tables for X-ray Crystallography, 1974, p. 71) and $\mathrm{O}^{2-}$ (Tokonami, 1965) scattering factors used and corrected for anomalous dispersion (International Tables for X-ray Crystallography, 1974, p. 148). Patterson synthesis used to determine positions of heavy atoms. Oxygens located in Fourier synthesis, using SHELX76 (Sheldrick, 1976). Least-squares refinement on $F, w=1 / \sigma^{2}$, including isotropic secondary extinction performed with program LINEX (Becker \& Coppens, 1975). Best results obtained for isotropic extinction with a prevalence of mosaic spread (type I) over particle size (type II) and Lorentzian shape for distribution function ( $\eta=3 \cdot 6^{\prime \prime}$ and $r=4 \mu \mathrm{~m}$ ). Refinement with isotropic temperature

[^0]Table 1. Atomic positions and equivalent isotropic thermal coefficients

$$
B_{\mathrm{eq}}=\frac{4}{3} \sum_{l} \Sigma_{j} \beta_{l j} \mathbf{e}_{i} \cdot \mathbf{e}_{j} .
$$

|  | Wyckoff position | $x$ | $y$ | $z$ | $B_{\text {cq }}\left(\AA^{2}\right)$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Zr}(1)$ | 2(b) | 0 | 0 | 0 | 1.06 (3) |
| $\mathbf{Z r}(2)$ | 4(f) | $\frac{1}{3}$ | $\frac{2}{3}$ | 0.98132 (8) | 0.93 (2) |
| Mo | 12(i) | 0.32871 (7) | $0 \cdot 33891$ (7) | $0 \cdot 14900$ (3) | 0.80 (2) |
| O(1) | 12(i) | 0.1731 (6) | $0 \cdot 1664$ (6) | $0 \cdot 1008$ (4) | 1.9 (2) |
| O(2) | 12(i) | 0.3323 (9) | 0.4959 (6) | 0.0814 (3) | 1.9 (2) |
| $\mathrm{O}(3)$ | 12(i) | 0.4979 (5) | 0.3364 (9) | 0.1178 (3) | 1.7 (1) |
| O(4) | 12(i) | $0 \cdot 3189$ (6) | $0 \cdot 3522$ (6) | 0.2923 (3) | $2 \cdot 3$ (2) |

factors gave $R=0.097$; correction for isotropic extinction (type I) resulted in $R=0.063$. Refinement with anisotropic temperature factors gave $R=0.027$. Since Hamilton's (1965) ratio $\mathscr{R}_{27,708,0.005}=1.0536$ is much less than $R_{\text {iso }+ \text { ext }} / R_{\text {aniso }}(2 \cdot 33)$, the anisotropic model was justified at a confidence level of more than $95 \%$. The atomic and thermal parameters are given in Table 1. A last refinement was done with 855 reflexions ( $F_{\text {obs }}>\sigma$ ), which resulted in $R=0.031, w R=0.028$, $S=2.638, \quad(\Delta / \sigma)_{\max }=0.007, \quad g=1.30(3) \times 10^{-4}$, improving the standard deviations on the interatomic distances. The highest peak in a difference Fourier synthesis was $0.73 \mathrm{e} \AA^{-3}$ near $\mathrm{Zr}(1)$.*

Discussion. Zirconium environment. The $\mathbf{Z r}^{4+}$ cations occupy two sets of special equivalent positions of multiplicity two and four. They are at the centre of a trigonal antiprism which will be subsequently likened to a $\mathrm{ZrO}_{6}$ octahedron. $\mathrm{Zr}(1)$ with point symmetry $\overline{3}$ has six O (1)-atom neighbours. $\mathrm{Zr}(2)$ with point symmetry 3 has two sets of three O -atom neighbours. The average distance $\mathrm{Zr}-\mathrm{O}$ is $2.086 \AA$ which is in good agreement with the distance of $2.07 \AA$ predicted from Shannon's (1976) ionic radii. Bond lengths, bond angles and bond strengths are listed in Table 2.

Molybdenum environment. The $\mathrm{Mo}^{6+}$ cation is in a distorted tetrahedral arrangement. One oxygen $\mathrm{O}(4)$ is bonded only to molybdenum while the three other oxygens are shared by one zirconium and one molybdenum. As a consequence, there are two types of Mo-O distances: one unusual short Mo-O(4) distance of 1.690 (3) $\AA$ and three longer distances with an average value of $1.764 \AA$ for a predicted distance of $1.76 \AA$ (Shannon, 1976). Bond lengths, bond angles and bond strengths are listed in Table 2.

Crystal network. The structure is a two-dimensional network with layers parallel to (001) at $c / 2$ intervals. The $\mathrm{MoO}_{4}$ tetrahedra are connected to the $\mathrm{ZrO}_{6}$

[^1]octahedra via three monodentate common corners: $\mathrm{O}(1), \mathrm{O}(2), \mathrm{O}(3)$. The bonding along [001] between two close layers is of van der Waals type; it takes place through $12 \mathrm{O}(4)-\mathrm{O}(n)$ distances which lie between 3.01 and $4.04 \AA$ (Table 2). The $\mathrm{O}(4)$ coordination

Table 2. Bond lengths ( $\AA$ ), bond angles $\left({ }^{\circ}\right)$ and bond strengths (s)
$\mathrm{MoO}_{4}$ tetrahedra

| $s$ |  |  |  |  |
| ---: | :---: | :---: | :---: | ---: |
| Mond distribution |  |  |  |  |
| Mo(1) | $1.763(7)$ | 1.50 | $1 \sigma+0.5 \pi$ | $\mathrm{O}(1)-\mathrm{O}(2) 2.903(17)$ |
| $-\mathrm{O}(2)$ | $1.762(5)$ | 1.50 | $1 \sigma+0.5 \pi$ | $-\mathrm{O}(3) 2.860(16)$ |
| $-\mathrm{O}(3)$ | $1.767(5)$ | 1.48 | $1 \sigma+0.5 \pi$ | $-\mathrm{O}(4) 2.825(12)$ |
| $-\mathrm{O}(4)$ | $1.690(3)$ | 1.89 | $1 \sigma+1 \pi$ | $\mathrm{O}(2)-\mathrm{O}(2) 2.887(12)$ |
|  |  |  |  | $-\mathrm{O}(4) 2.836(8)$ |
|  |  |  |  | $\mathrm{O}(3)-\mathrm{O}(4) 2.790(9)$ |


| $\mathrm{O}(1)-\mathrm{Mo}-\mathrm{O}(2)$ | 110.89 (4) |  | $\mathrm{O}(2)-\mathrm{Mo}$ |
| :---: | :---: | :---: | :---: |
| $\mathrm{O}(1)-\mathrm{Mo}-\mathrm{O}(3)$ | 108.26 (3) |  | $\mathrm{O}(2)-\mathrm{Mo}$ |
| $\mathrm{O}(1)-\mathrm{Mo}-\mathrm{O}(4)$ | 109.74 (4) |  | $\mathrm{O}(3)-\mathrm{Mo}$ |
| $\mathrm{ZrO}_{6}$ polyhedra |  |  |  |
| $\mathrm{Zr}(1)-\mathrm{O}\left(1^{\text {to }}\right.$ vi $)$ | (6x) | 2.088 (8) | $\stackrel{s}{0.68}$ |
| $\mathrm{Zr}(2)-\mathrm{O}\left(2^{1.1 \times, x}\right)$ | (3x) | 2.087 (8) | 0.68 |
| $\mathrm{O}\left(3^{\text {r,viluvili }}\right)$ | (3x) | 2.081 (7) | $0 \cdot 69$ |

$\mathrm{O}\left(1^{\prime}\right)-\mathrm{Zr}(1)-\mathrm{O}\left(1^{11}\right) \quad(6 \times) 91 \cdot 19$ (3) $\mathrm{O}\left(3^{\text {vil }}\right)-\mathrm{Zr}(2)-\mathrm{O}\left(3^{\text {vi }}\right)(3 \times) 91.91$ (3)
$\mathrm{O}\left(1^{1}\right)-\mathrm{Zr}(1)-\mathrm{O}\left(1^{4}\right) \quad$ ( $\left.6 \times \mathrm{x}\right) 88.81$ (3) $\mathrm{O}\left(2^{\text {l }}\right)-\mathrm{Zr}(2)-\mathrm{O}\left(3^{\text {viI }}\right)(3 \times) 87.92$ (3) $\mathrm{O}\left(2^{\prime}\right)-\mathrm{Zr}(2)-\mathrm{O}\left(2^{\times 1}\right) \quad(3 \times) 91.54$ (3) $\mathrm{O}\left(2^{\prime}\right)-\mathrm{Zr}(2)-\mathrm{O}\left(3^{\mathrm{v}}\right) \quad$ (3x) 88.62 (3)
$O(4)$ coordination polyhedron

| $\mathrm{O}(4)-\mathrm{O}\left(1^{\times 1}\right)$ | 3.99 (1) | $\mathrm{O}(4)-\mathrm{O}\left(1^{\text {xil }}\right.$ ) | 3.29 (2)** |
| :---: | :---: | :---: | :---: |
| -O(2x) | 3.88 (1) | -O(1xii) | 3.13 (2)* |
| -O(3*) | 4.06 (1) | -O(2xii) | 3.01 (1)* |
| -O(4xi) | 4.03 (2) | -O(2 ${ }^{\text {ivi }}$ ) | 3.15 (2)* |
| -O(4xill | 3.06 (1) | -O(3xi) | 3.41 (1)* |
| -O(4 ${ }^{\text {xiV }}$ ) | 3.48 (2) | -O(3xiv) | 3.21 (2)* |

Symmetry code: (i) $x, y, z$; (ii) $-y, x-y, z$; (iii) $y-x,-x, z$; (iv) $-x$, $-y,-z$; (v) $y, y-x,-z$; (vi) $x-y, x,-z$; (vii) $1-x, 1-y,-z$; (viii) $y$, $1+y-x,-z$; (ix) $1-y, 1+x-y, z ;(\mathrm{x}) y-x, 1-x, z$; (xi) $y, x, \frac{1}{2}+z$; (xii) $x, x-y, \frac{1}{2}-z$; (xiii) $y-x, y, \frac{1}{2}-z$; (xiv) $1-y, 1-x, \frac{1}{2}-z$.


Fig. 1. Projection of the $\mathrm{HT} \mathrm{Zr}\left(\mathrm{MoO}_{4}\right)_{2}$ structure along the $c$ axis.
polyhedron can be visualized as a strongly distorted Archimedean cuboctahedron, which is the coordination polyhedron of an anion with respect to its nearest-neighbour anions in ideal cubic close packing. The pseudo-hexagon with six oxygen atoms, asterisked in Table 2, is nearly parallel to (001). A projection of the $\mathrm{HT} \mathrm{Zr}\left(\mathrm{MoO}_{4}\right)_{2}$ structure along the $c$ axis is given in Fig. 1.

The outstanding feature of this structure is the oxygen $O(4)$ bonded only to molybdenum. As a result, the Mo-O(4) bond strength (Zachariasen, 1978) has a much higher value than that of the other three (Table 2) and the $O(4)$ average quadratic vibration is 1.5 times lower along the $z$ axis than in the (001) plane. According to Schröder (1975) this structural deformation goes along with a modification of the $\pi$-bond distribution around the Mo atom, which gives to the Mo-O(4) bond a pronounced covalent character (Table 2). This structure, common to $\mathrm{Hf}\left(\mathrm{MoO}_{4}\right)_{2}$ and $\mathrm{HT} \mathrm{Zr}\left(\mathrm{MoO}_{4}\right)_{2}$, is unique since, to our knowledge, there is no other isostructural $A\left(\mathrm{XO}_{4}\right)_{2}$ compound. However, the structural unit made up of $\mathrm{ZrO}_{6}$ octahedra surrounded by six monodentate $\mathrm{MoO}_{4}$ tetrahedra, alternately pointing up and down, is very similar in configuration to several structures that have tetrahedral anions, such as the glaserite $\mathrm{K}_{3} \mathrm{Na}\left(\mathrm{SO}_{4}\right)_{2}$ structure (Moore, 1973).

The differences between the structure reported by Rimsky, Thoret \& Freundlich (1968) and ours lie in the Wyckoff position of the $M^{\text {IV }}(1)$ cation [2(a) instead of $2(b)]$ and in the orientation of the $\mathrm{MoO}_{4}$ tetrahedra, which led them to a three-dimensional network and a twelvefold-coordinated $M^{\text {iv }}(1)$ cation.

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# Structure du Sulfure d'Europium et de Bismuth $\mathbf{E u}_{1,1} \mathbf{B i}_{\mathbf{2}} \mathbf{S}_{\mathbf{4}}$ 

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#### Abstract

M_{r}=698 \cdot 18\), hexagonal, $P 6_{3} / m, \quad a=$ 24.820 (8), $\quad c=4.080$ (1) $\AA, \quad V=2176.7(1.9) \AA^{3}, Z$ $=12, D_{m}=6 \cdot 2(1), D_{x}=6 \cdot 39(1) \mathrm{Mg} \mathrm{m}^{-3}, \lambda($ Mo $K \bar{\alpha})$ $=0.71069 \AA, \quad \mu=55.5 \mathrm{~mm}^{-1}, \quad F(000)=3516, \quad T=$ $293 \mathrm{~K}, R=0.053$ for 1216 independent observed reflections. The Eu atoms have eightfold prismatic coordination [trigonal prism (Eu-S: 3.02 to $3.17 \AA$ ) with Eu contacts to two other S atoms in the equatorial plane of the prism (Eu-S: $3 \cdot 10$ to $3 \cdot 26 \AA$ )] and the Bi atoms have sixfold octahedral coordination ( $\mathrm{Bi}-\mathrm{S}$ : 2.52 to $3.63 \AA$ ). There are large channels around the sixfold axes, which are not completely empty and contain two six-coordinated sites partially occupied by europium.


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Introduction. L'étude du système $\mathrm{E}_{1} \mathrm{~S}_{4}-\mathrm{Bi}_{2} \mathrm{~S}_{3}$ nous a permis de décrire le composé $\mathrm{Eu}_{2} \mathrm{BiS}_{4}$ où l'europium présente deux états de valence (Lemoine, Carré \& Guittard, 1982). L'étude du système EuS- $\mathrm{Bi}_{2} \mathrm{~S}_{3}$ nous conduit au composé $E u_{1, \mathrm{I}} \mathrm{Bi}_{2} \mathrm{~S}_{4}$ dans lequel l'europium possède la valence 2 ét qui présente un caractère semi-métallique. Des monocristaux de cette nouvelle phase ont été isolés dans une préparation obtenue par mélange de $\mathrm{Bi}_{2} \mathrm{~S}_{3}+\mathrm{EuS}$, en ampoule de silice scellée sous vide, chauffee à 1073 K et refroidie lentement. On peut également l'obtenir à partir du mélange $\mathrm{EuS}+2 \mathrm{Bi}$ (en poudre) +3 S , en ampoule de silice scellée sous vide et chauffée lentement à 1073 K ou encore par sulfuration dans un courant de sulfure d'hydrogène du


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[^1]:    * Lists of structure factors and anisotropic thermal parameters have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 42609 ( 6 pp .). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH 1 2HU, England.

